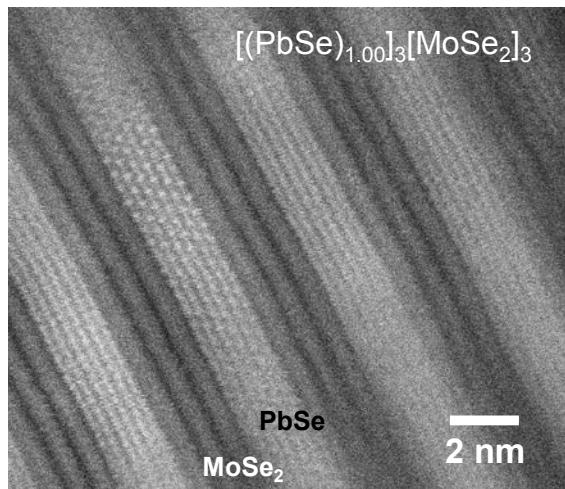
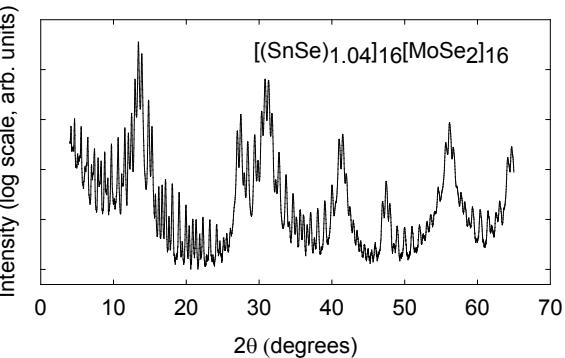


# *Trends in Thermoelectric Properties with Nanostructure: Ferecrysrtals with Designed Nanoarchitecture*

David C. Johnson

Matt Beekman,<sup>1</sup> Ryan Atkins,<sup>1</sup> Daniel B. Moore,<sup>1</sup> Michael D. Anderson,<sup>1,2</sup> Corinna Grosse,<sup>1</sup> Mary Smeller,<sup>1</sup> Colby Heideman,<sup>1</sup> Qiyin Lin,<sup>1</sup> Ngoc Nguyen,<sup>1</sup> Brian Lessig,<sup>1</sup> Ian M. Anderson,<sup>2</sup> Wolfgang Neumann,<sup>3</sup> Paul Zschack,<sup>4</sup>



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<sup>2</sup>Surface and Microanalysis Science Division, National Institute of Standards and Technology, Gaithersburg, MD

<sup>3</sup>Department of Physics, Humboldt University, Berlin

<sup>4</sup>Advanced Photon Source, Argonne National Laboratory, Argonne, IL



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OREGON NANOSCIENCE AND  
MICROTECHNOLOGIES INSTITUTE



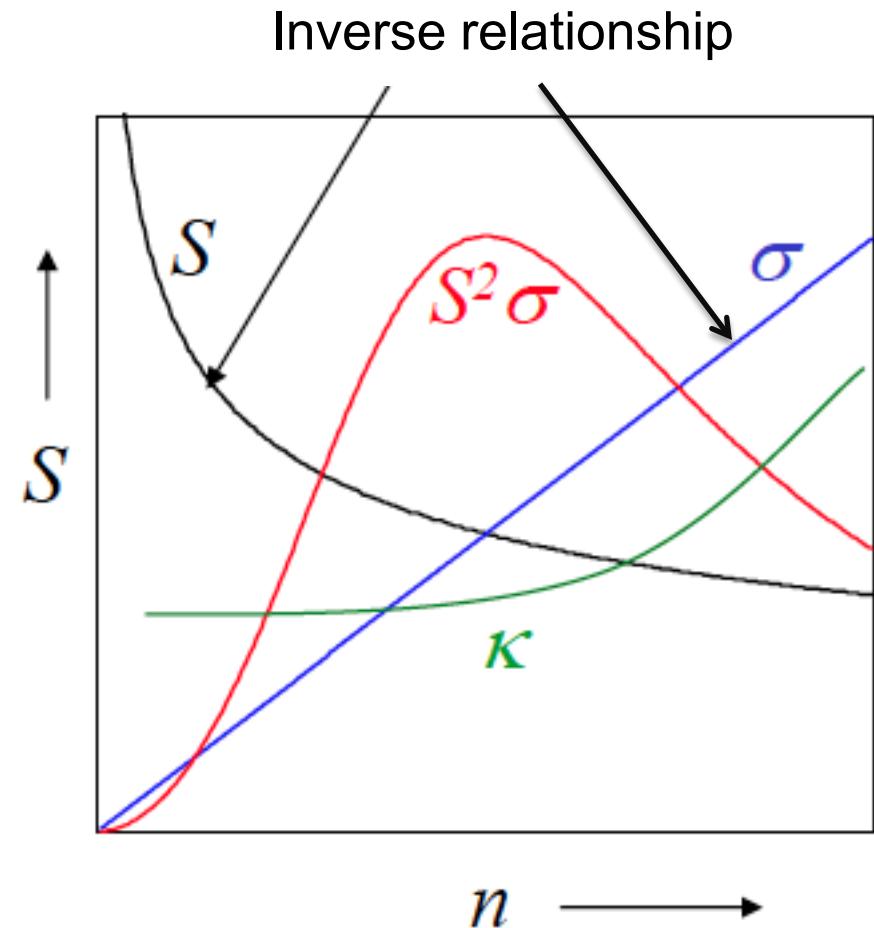
# Interrelationships Limit ZT

$$ZT = S^2 \sigma T / \kappa$$

$$S = S(n)$$

$$\sigma = n e \mu$$

$$\kappa = \kappa_{\text{lattice}} + \kappa_{\text{electronic}}$$

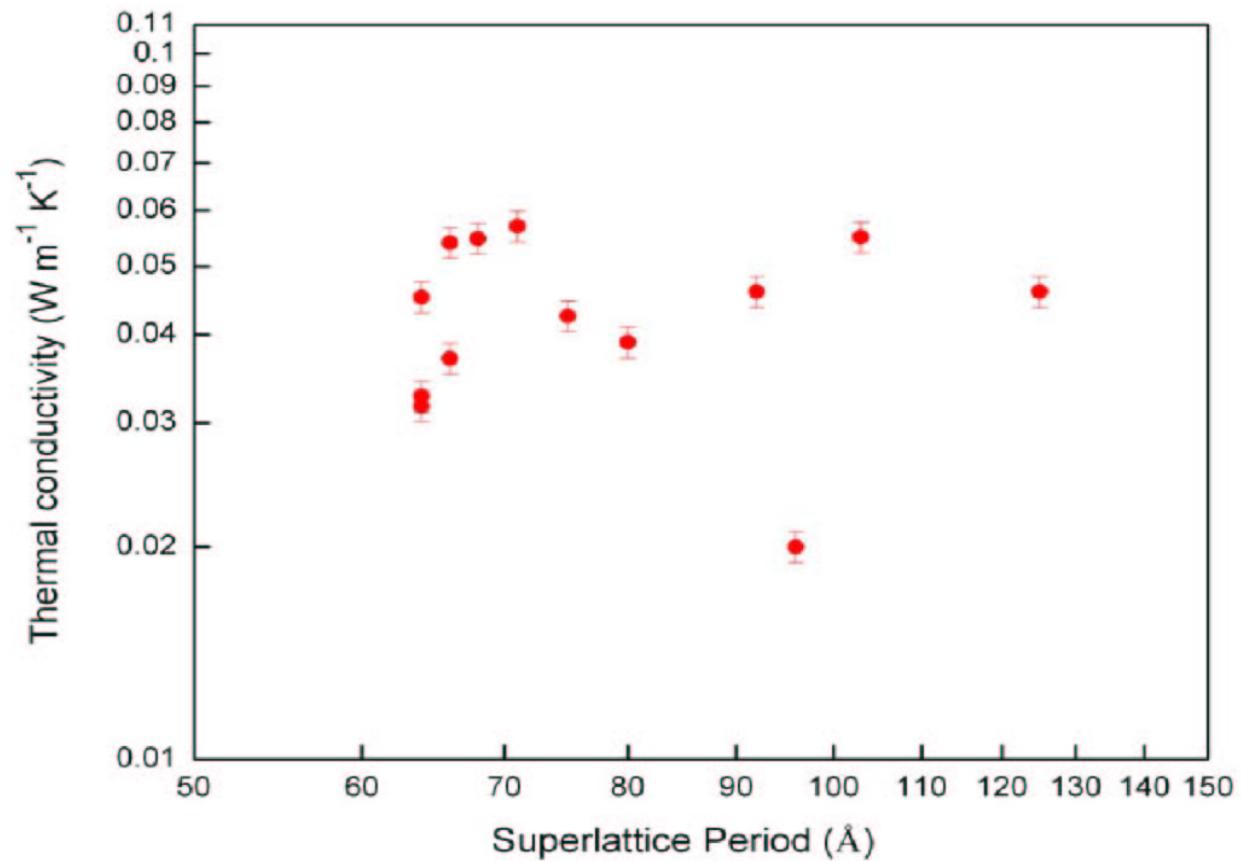
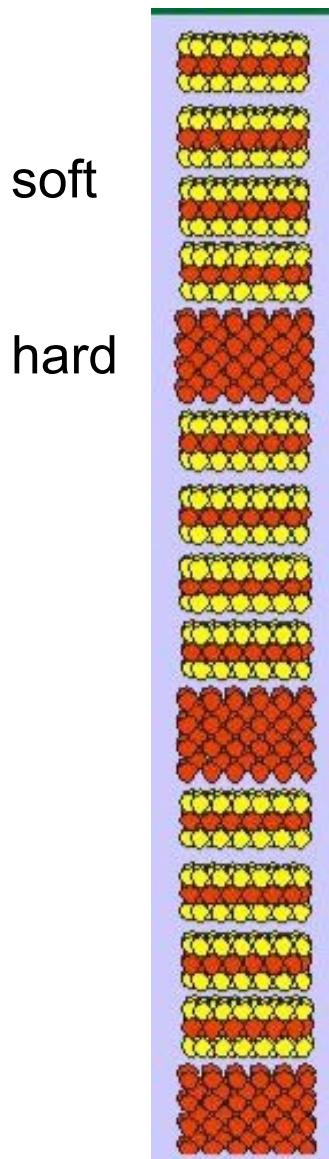


How do we design material with low  $\kappa_{\text{lattice}}$ ?



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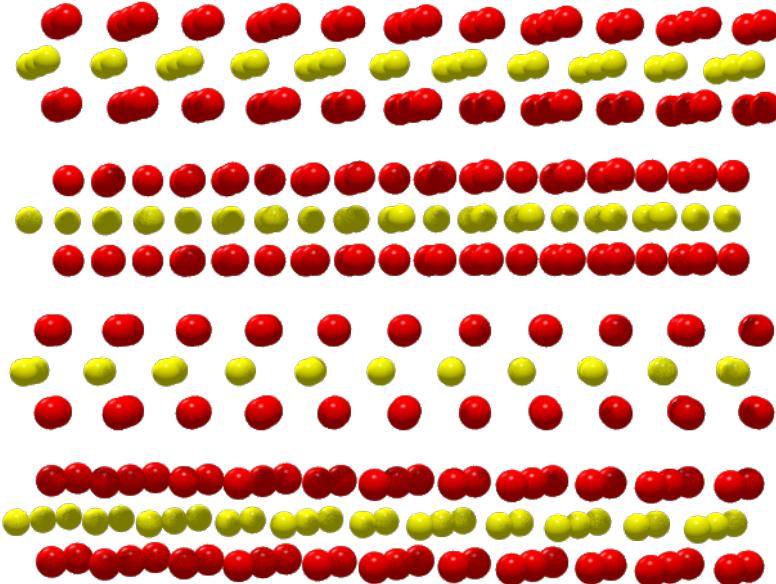
# Thermal Conductivity of $(W)_x(WSe_2)_y$



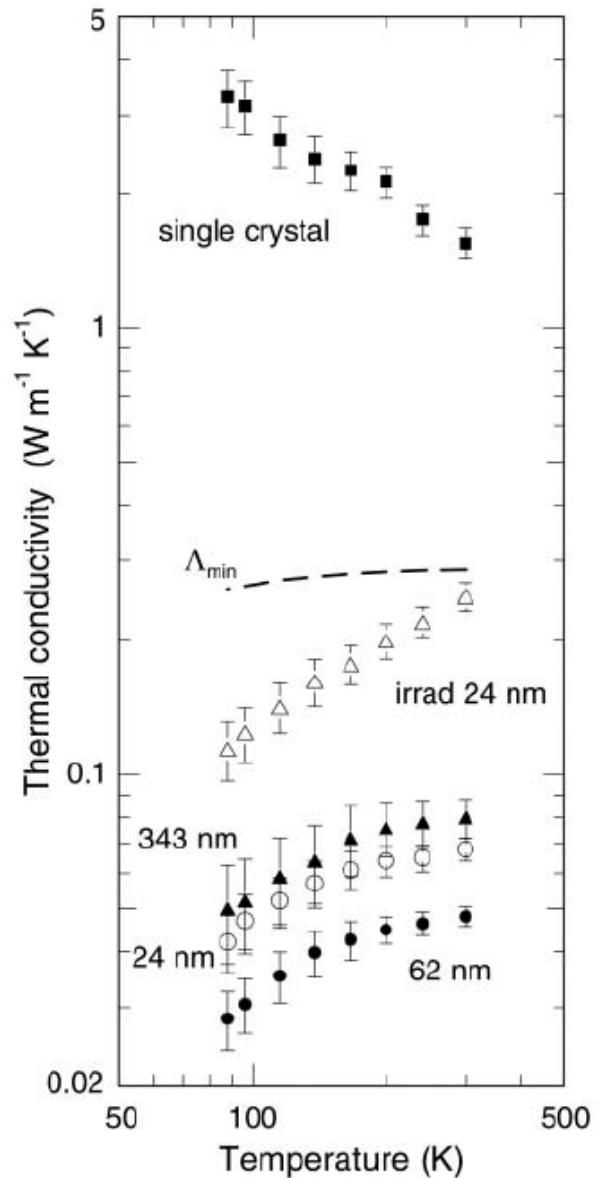
Lowest thermal conductivity ever measured  
for a fully dense solid

# Ultralow Thermal Conductivity in WSe<sub>2</sub>

- Crystalline within sheets
- Disordered between sheets

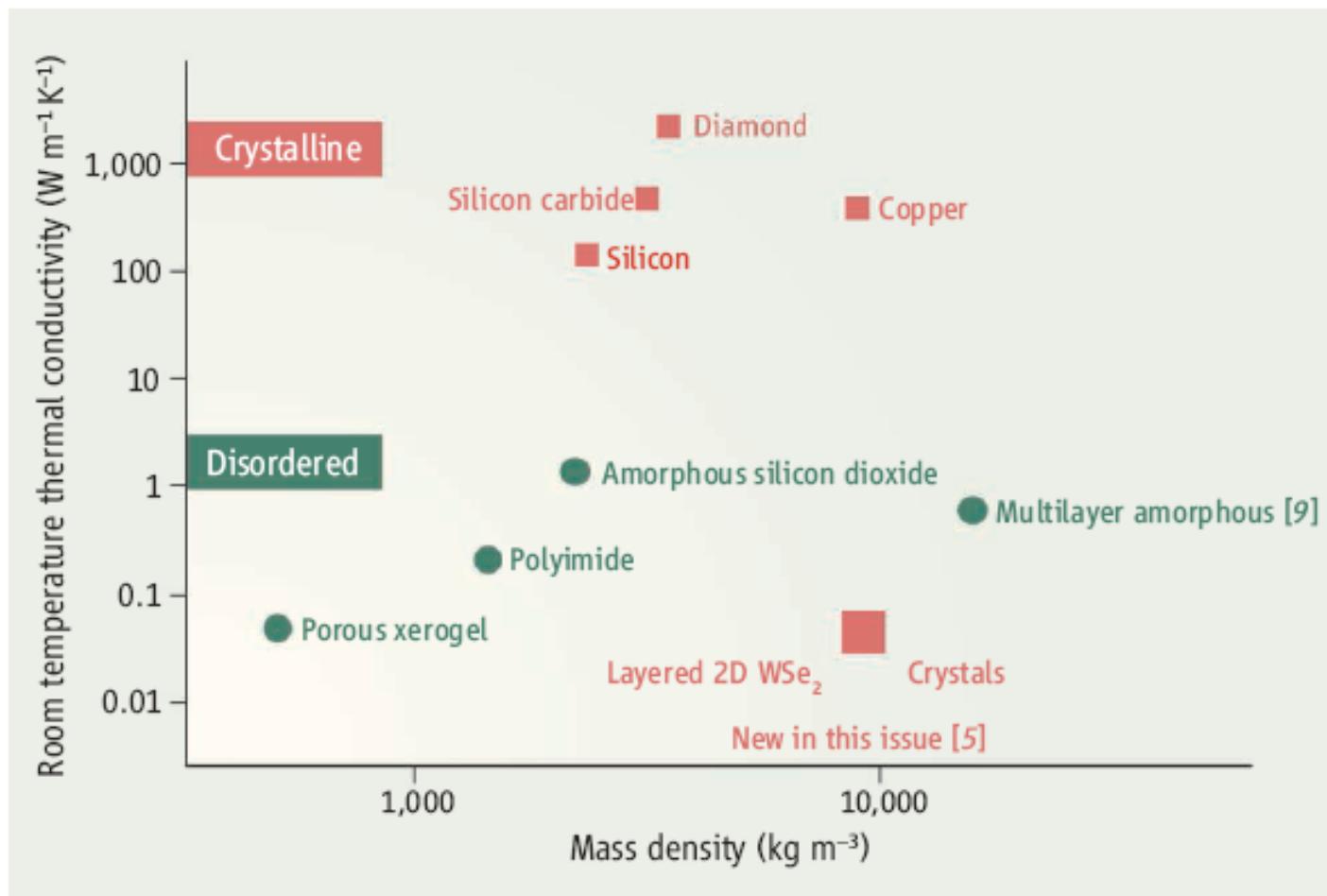


C. Chiritescu, D. G. Cahill, N. Nguyen, D. C. Johnson, A. Bodapati, P. Keblinski, P. Zschack, *Science* **2007**, *315*, 351.



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# New Lower Limit



K. E. Goodson, *Science* **2007**, 315, 342.

# Where else might this be possible?

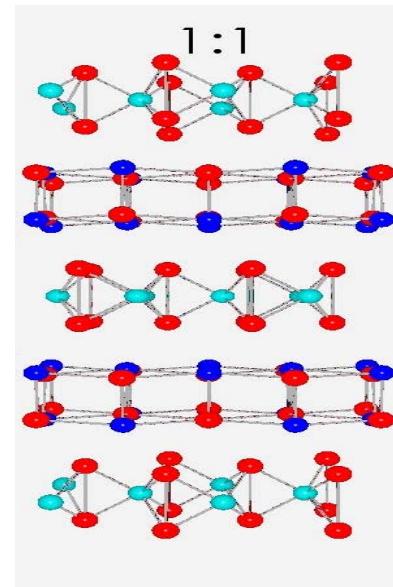
- Lattice mismatch in the a-b plane  $[(AX)_{1+y}]_n(MX_2)_m$
- $n,m = 1$  forms by direct reaction of the elements at  $\sim 1000^\circ\text{C}$  for  $\sim 30$  compounds – mostly sulfides
- Compounds can be metallic, semiconducting or insulating (magnetic, superconducting, etc. )
- $(\text{YbS})_{1.24}\text{NbS}_2$        $S = 60 \mu\text{V/K}$        $\rho = 1.4 \times 10^{-5} \Omega \cdot \text{m}$   
 $K = 0.8 \text{ W m}^{-1} \text{ K}^{-1}$        $ZT = 0.1$

$(\text{LnS})_m(\text{NbS}_2)_n$  and  $(\text{SnS})_m(\text{TiS}_2)_n$   $n = 1, 2$   $m = 1$   
power factor,  $6 \sim 12 \times 10^{-4} \text{ W/m} \cdot \text{K}^2$  at 300K  
 $ZT \sim .1 \text{-.3}$  if the  $K$  value is  $\sim 1 \text{ W m}^{-1} \text{ K}^{-1}$ ,

Miyazaki, Ogawa, Kajitani, Jap. J. Appl. Physics, 43 (2004) L1202

- $(\text{SnS})_1(\text{TiS}_2)_2$        $K_{\text{lattice}} \sim 0.8 \text{ W m}^{-1} \text{ K}^{-1}$ ,  
 $ZT \sim .4$  at  $400^\circ \text{ C}$

C. Wan, Y. Wang, N. Wang and K. Koumoto - ICT 2009 Freiburg



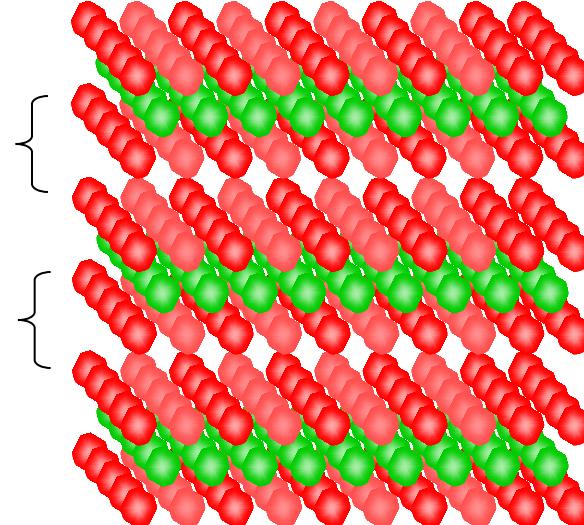
# Chalcogenide intergrowth compounds

Transition metal dichalcogenide



(e.g.  $\text{TiSe}_2$ ,  $\text{NbSe}_2$ , etc.)

van der Waals  
gap



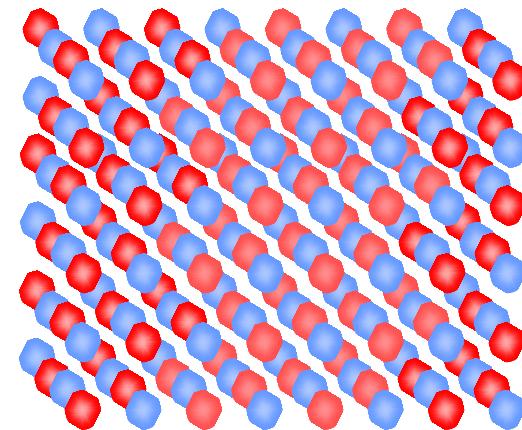
van der Waals  
gap

NaCl-like monochalcogenide



(e.g.  $\text{PbSe}$ ,  $\text{EuSe}$ , etc.)

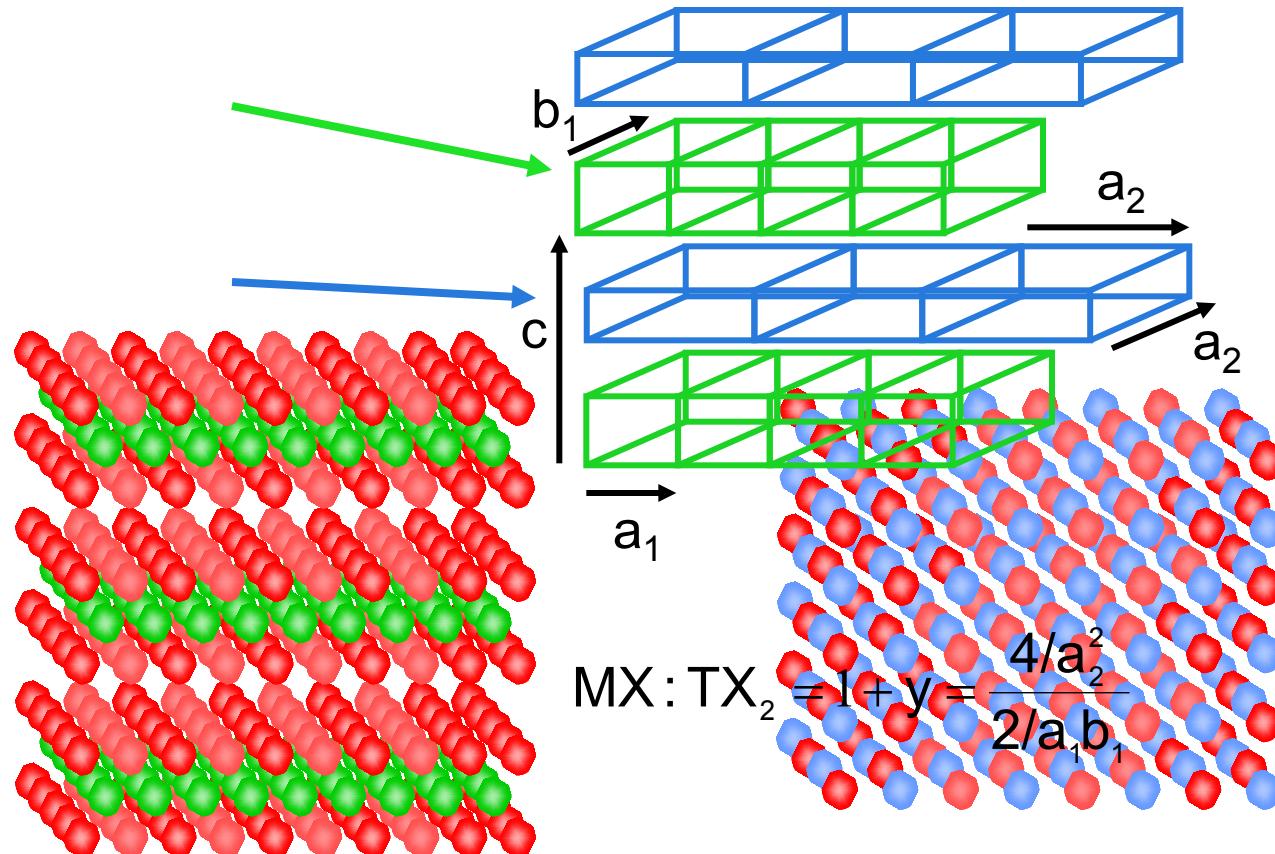
T  
X  
M



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# Chalcogenide intergrowth compounds

$[(MX)_{1+y}]_m [TX_2]_n$  ( $m, n = 1$  or  $2$ )



Review articles:

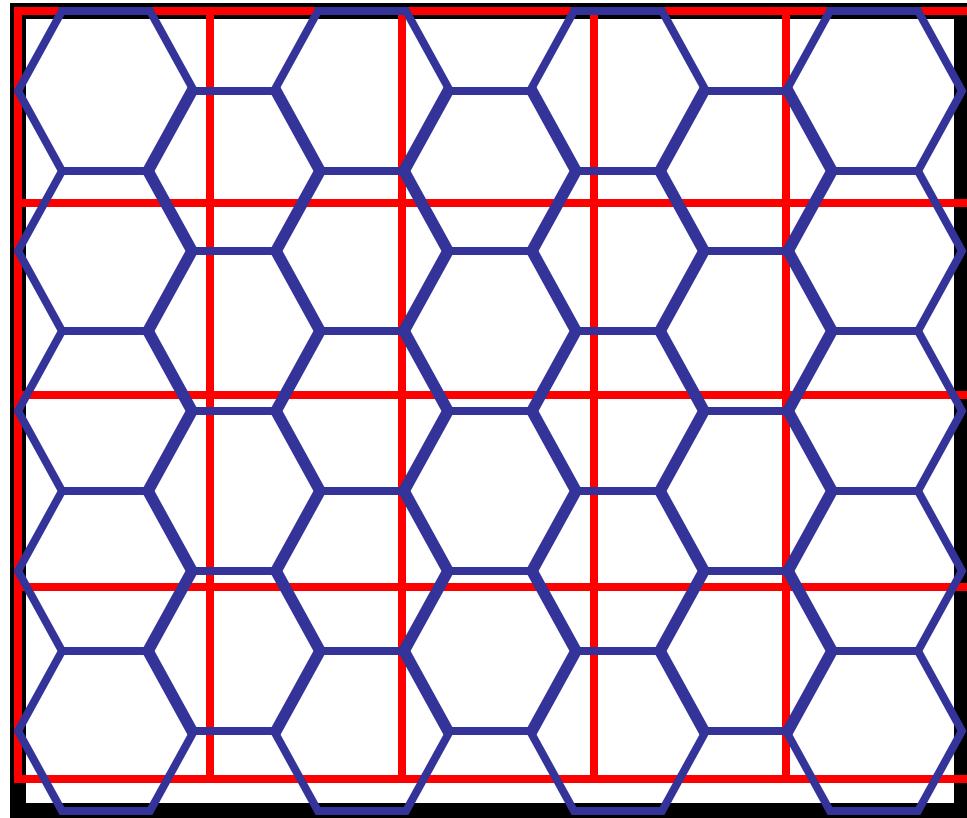
G.A. Wiegers, Prog. Solid State Chem. **24**, 1 (1996).

A. Meerchaert, Curr. Opin. Sol. State Mater. Sci. **1**, 25 (1996).



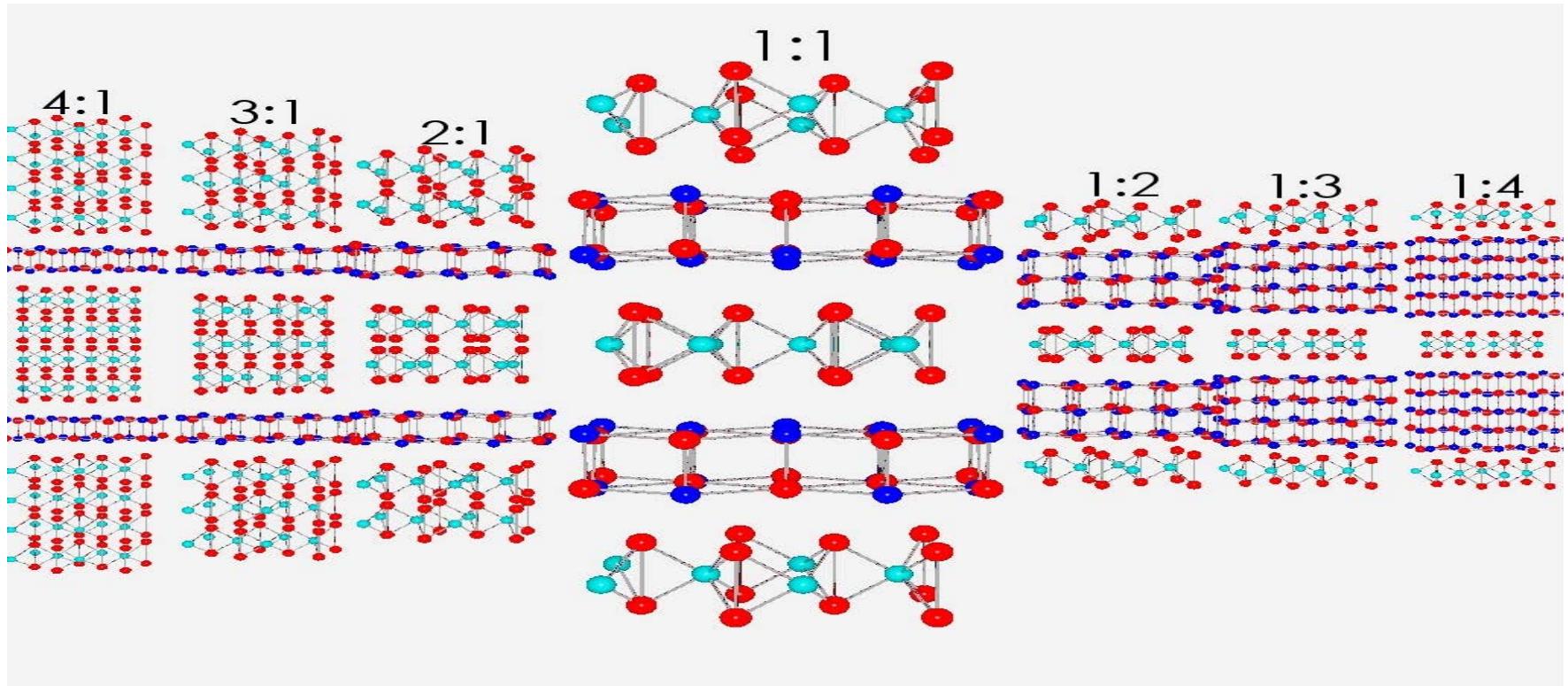
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# Different atom densities in x-y plane – no epitaxy



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# Controlled Nanostructure



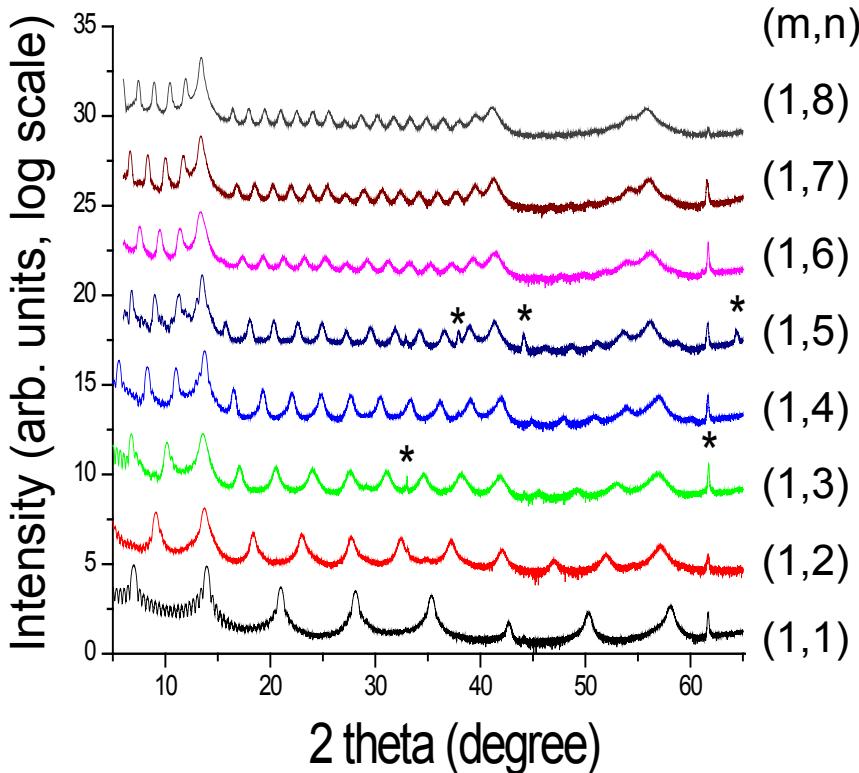
If the 1:1 structure is stable, all local bonding is satisfied in any compound,  $m:n$ . Reasonable to assume that all  $m:n$  compounds should be at least metastable.



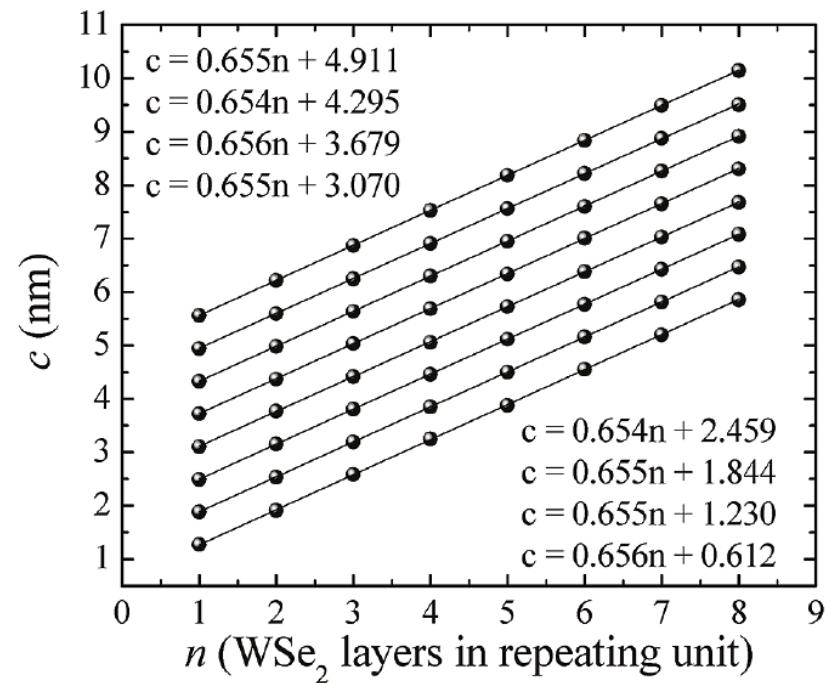
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# Entire families of compounds can be prepared

Only (00*l*) reflections are observed  
(\* indicates sample holder or substrate)



c-axis repeat thickness



First 64 members of  $[(\text{PbSe})_{0.99}]_m[\text{WSe}_2]_n$  prepared

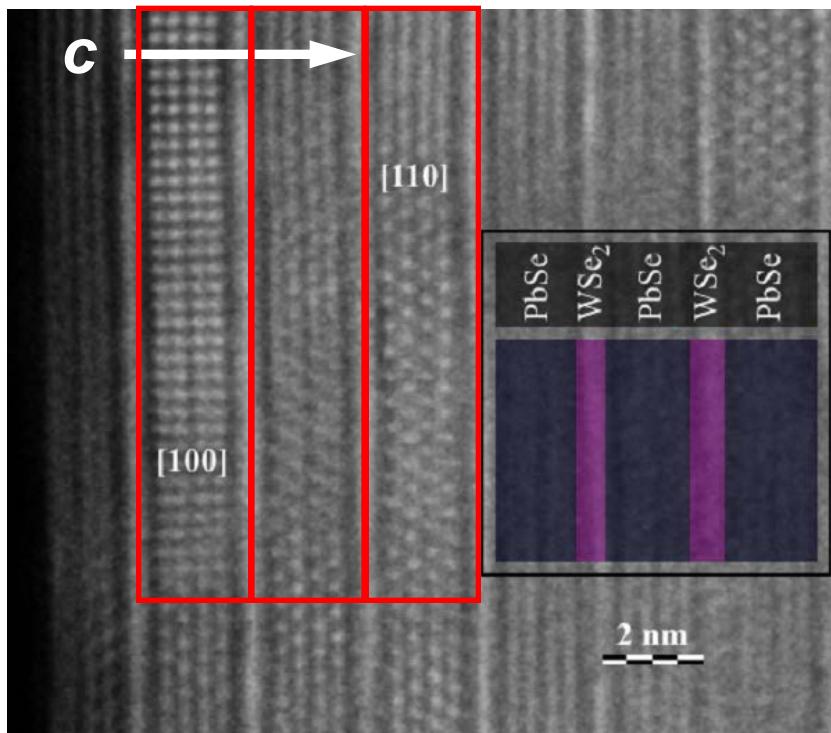
Q. Lin et al., Chem. Mater. **22**, 1002 (2010).



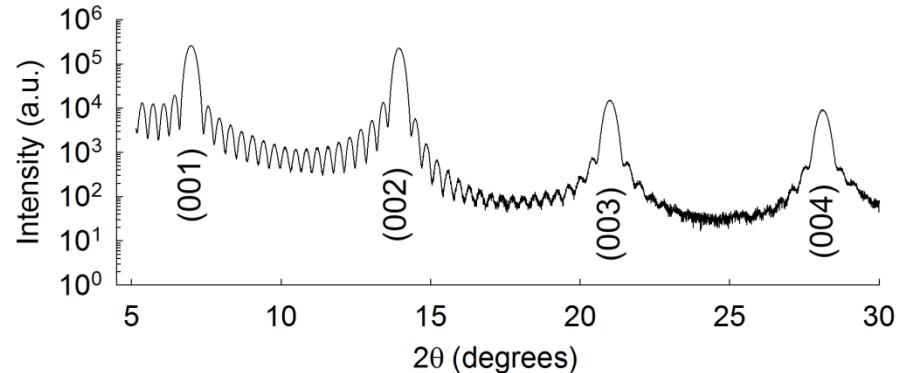
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# Precisely intergrown layers lacking interlayer registry

STEM image from  
[(PbSe)<sub>0.99</sub>]<sub>2</sub>[WSe<sub>2</sub>]<sub>1</sub>



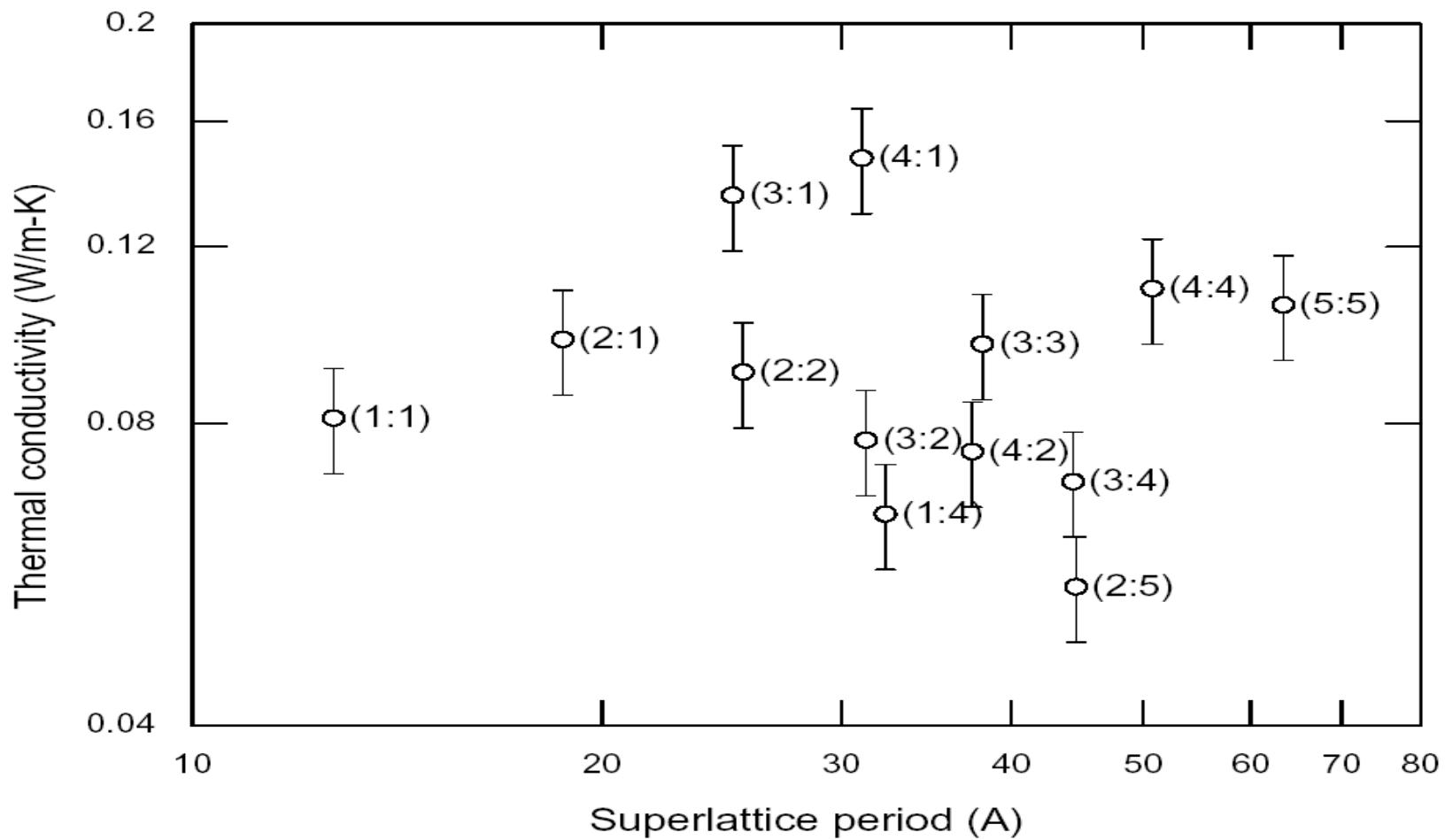
XRR from [(PbSe)<sub>0.99</sub>]<sub>2</sub>[WSe<sub>2</sub>]<sub>1</sub>



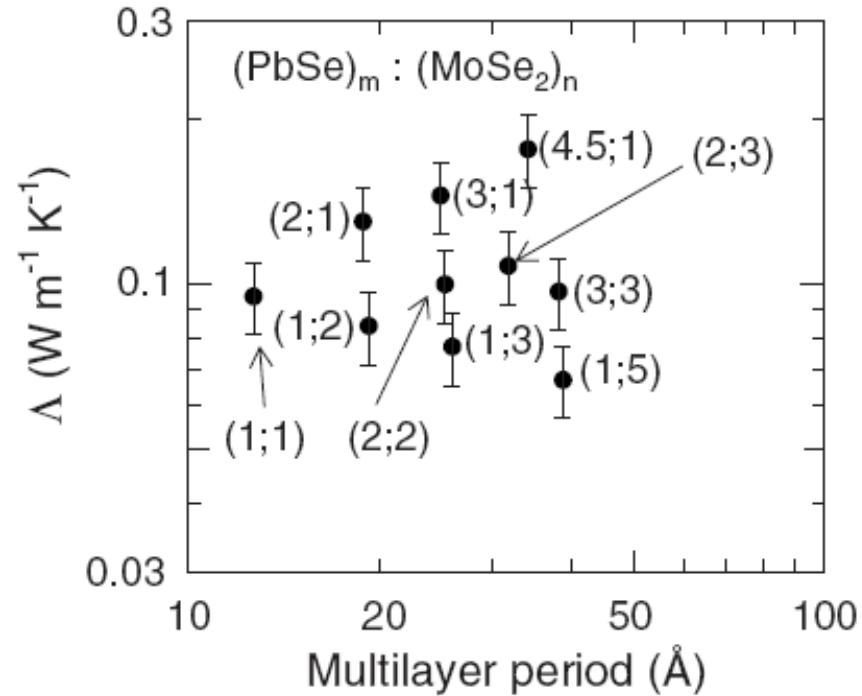
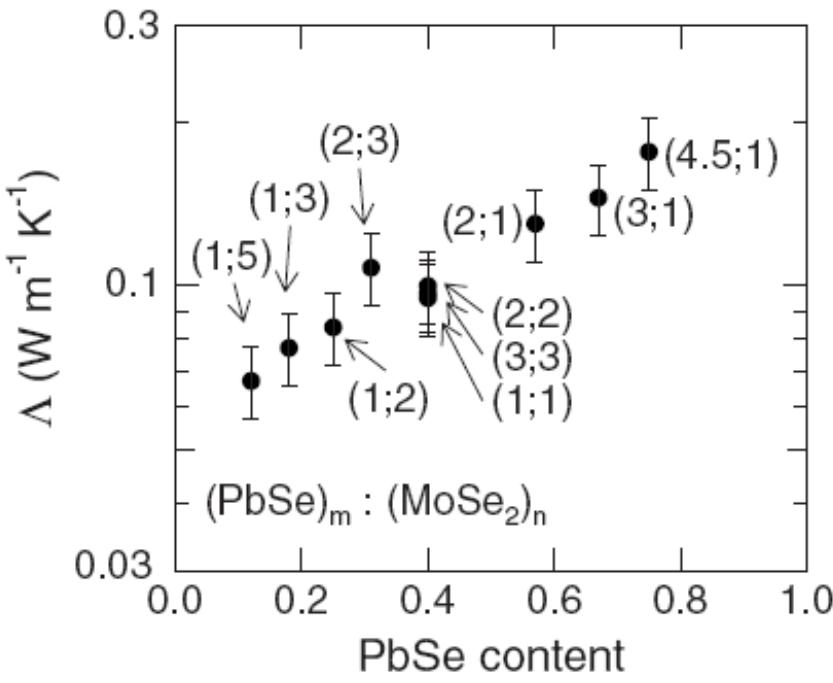
- ◆ In-plane crystallinity; crystallographically aligned components
- ◆ Atomically abrupt interfaces
- ◆ Precisely layered constituents
- ◆ Layer-to-layer mis-registration: similar to turbostratic disorder

Ferecystals – from Latin - almost

# Thermal Conductivity $[(\text{PbSe})_{0.99}]_n(\text{WSe}_2)_m$



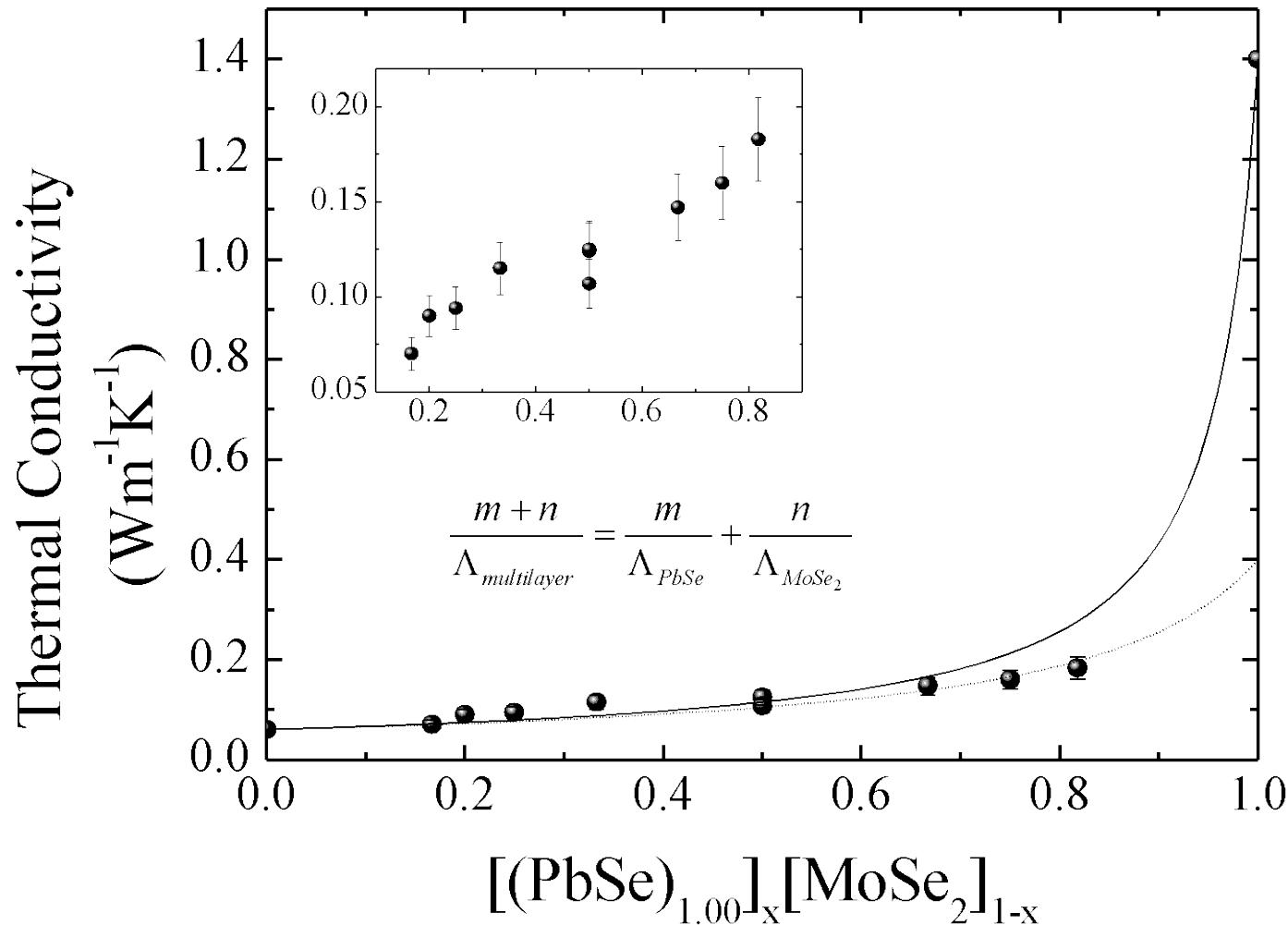
# Ultra-low cross-plane thermal conductivity



Thermoreflectance measurements:

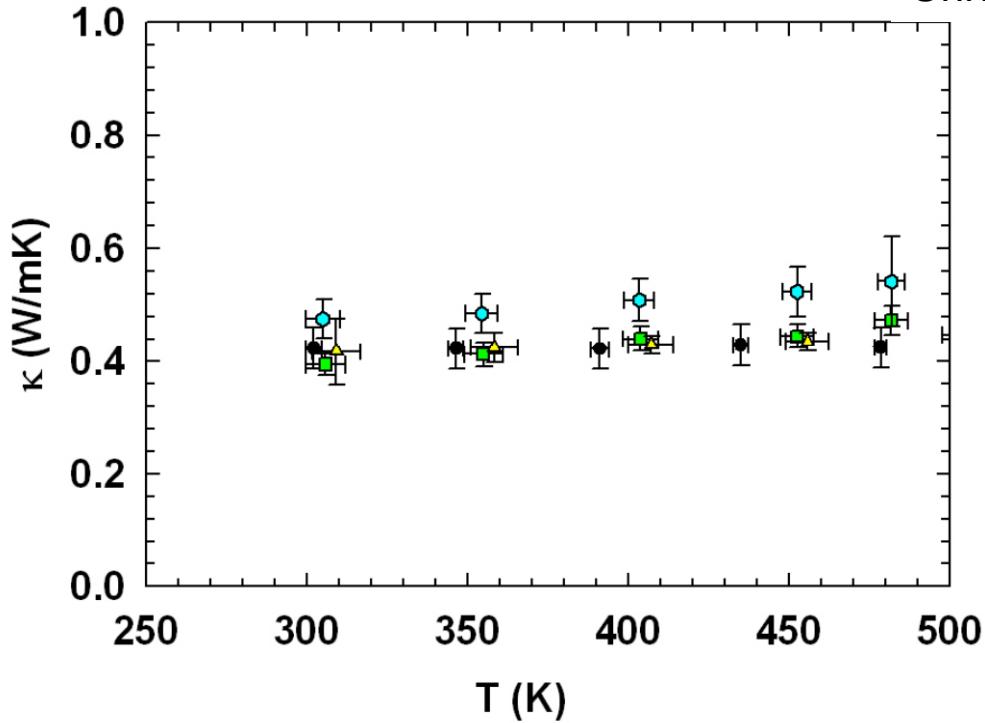
C. Chiritescu and Prof. D.G. Cahill, University of Illinois at Urbana-Champaign

# Cross Plane Thermal Conductivity – Sum of Constituents

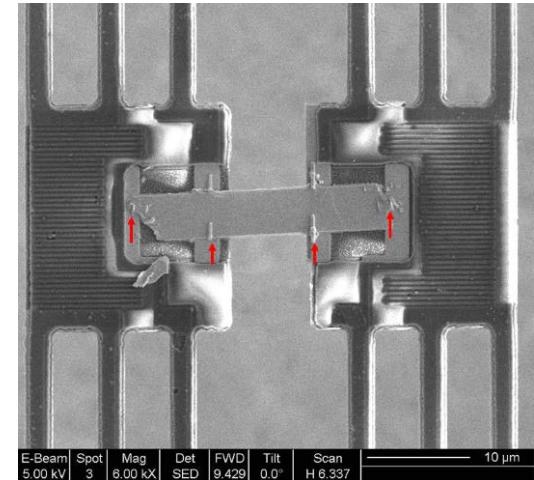


# In plane Thermal Conductivity

In plane thermal conductivity measurements:  
A. Mavrokefalos and Professor Li Shi  
University of Texas at Austin



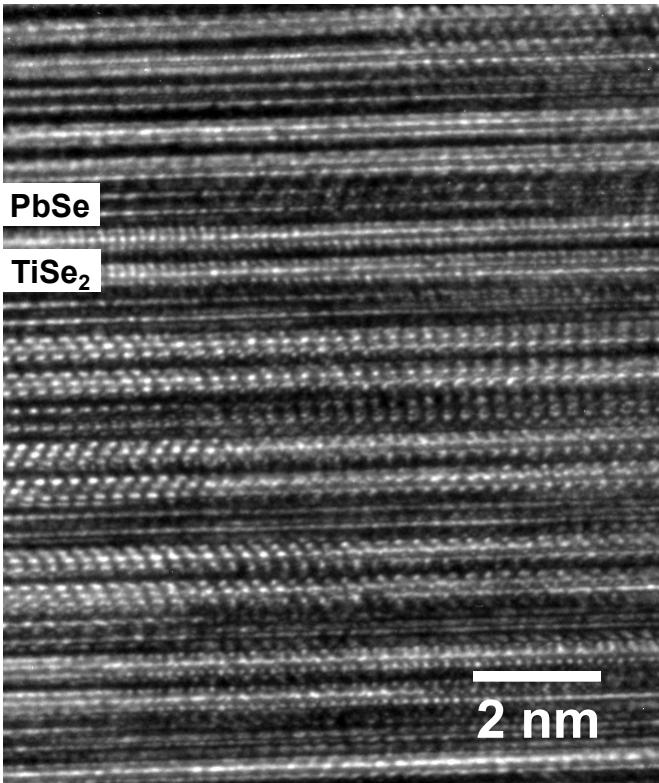
- (PbSe)<sub>2</sub>(WSe<sub>2</sub>)<sub>2</sub> before annealing
- (PbSe)<sub>3</sub>(WSe<sub>2</sub>)<sub>3</sub> after annealing
- ▲ (PbSe)<sub>4</sub>(WSe<sub>2</sub>)<sub>4</sub> after annealing
- (PbSe)<sub>2</sub>(WSe<sub>2</sub>)<sub>2</sub> after annealing



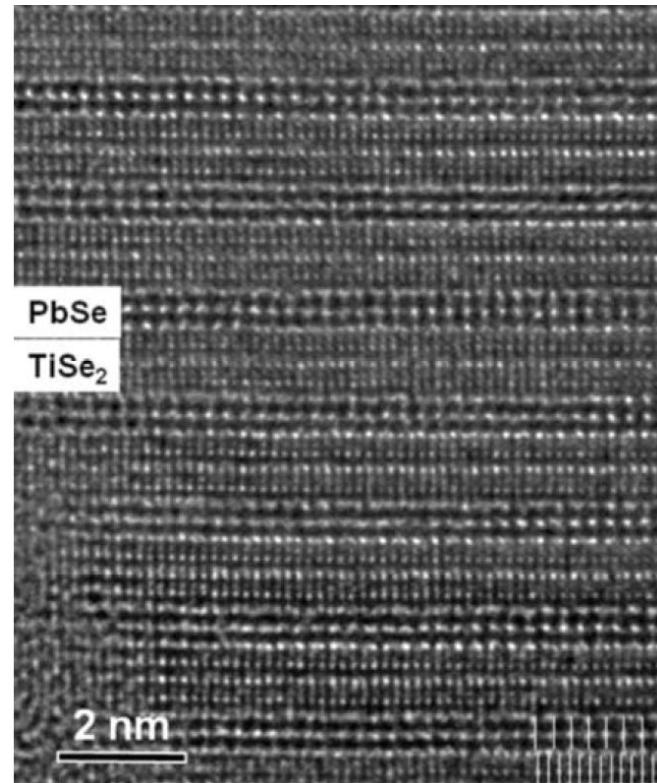
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# Precisely intergrown layers lacking interlayer registry

HRTEM image of  
 $[(\text{PbSe})_{1.16}]_1[\text{TiSe}_2]_2$ :



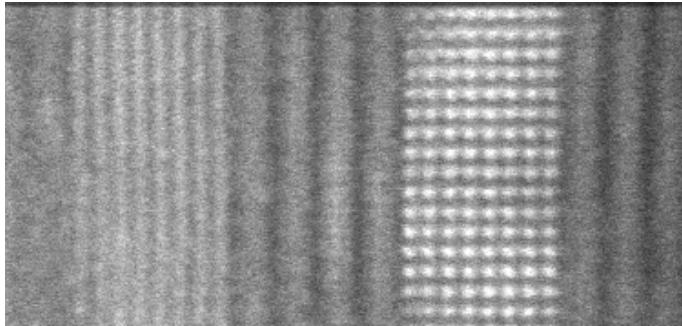
$[(\text{PbSe})_{1.16}]_1[\text{TiSe}_2]_2$   
Synthesized by direct  
reaction of the elements:<sup>1</sup>



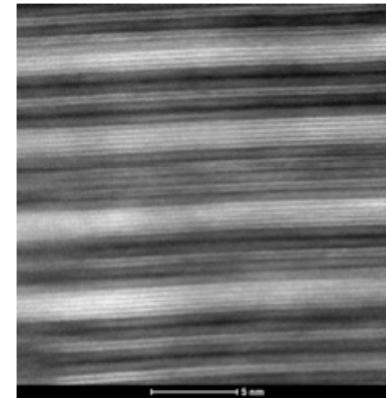
<sup>1</sup>N. Giang et al., *Phys. Rev. B* 82, 7 (2010),

# Isomers with controlled nanostructure

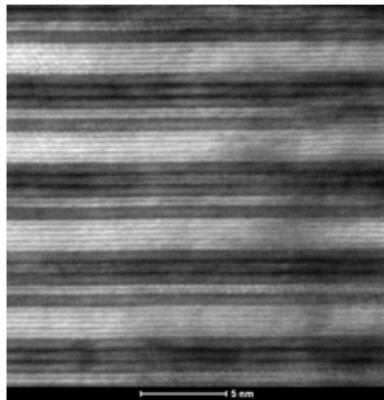
Four  $(\text{PbSe})_4(\text{NbSe}_2)_4$  Isomers



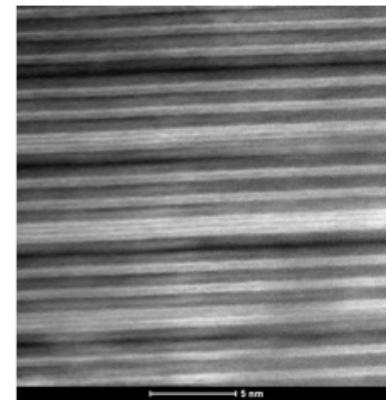
**4,4,0,0,0,0**



**2,3,2,1,0,0**



**3,3,1,1,0,0**

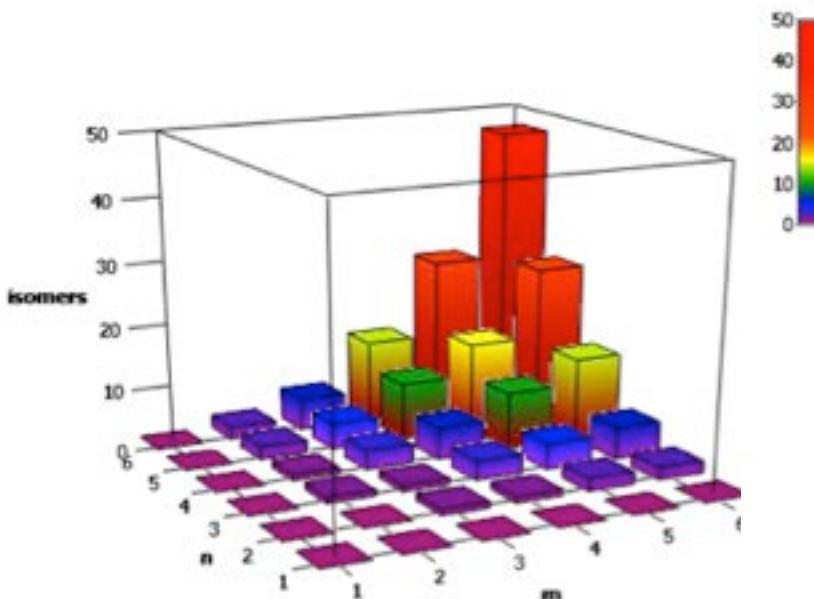


**2,2,1,1,1,1**



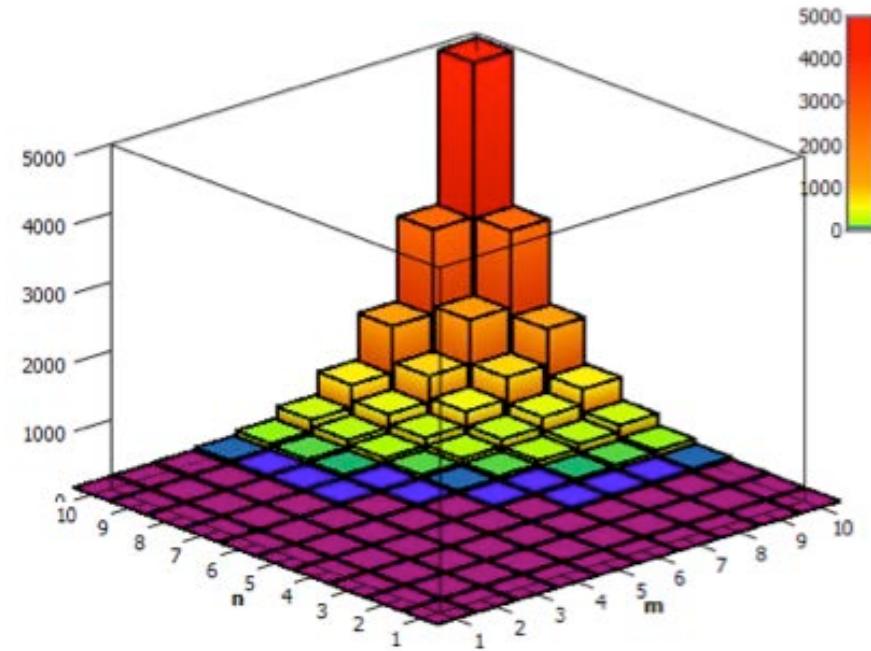
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# Many possible isomers ...



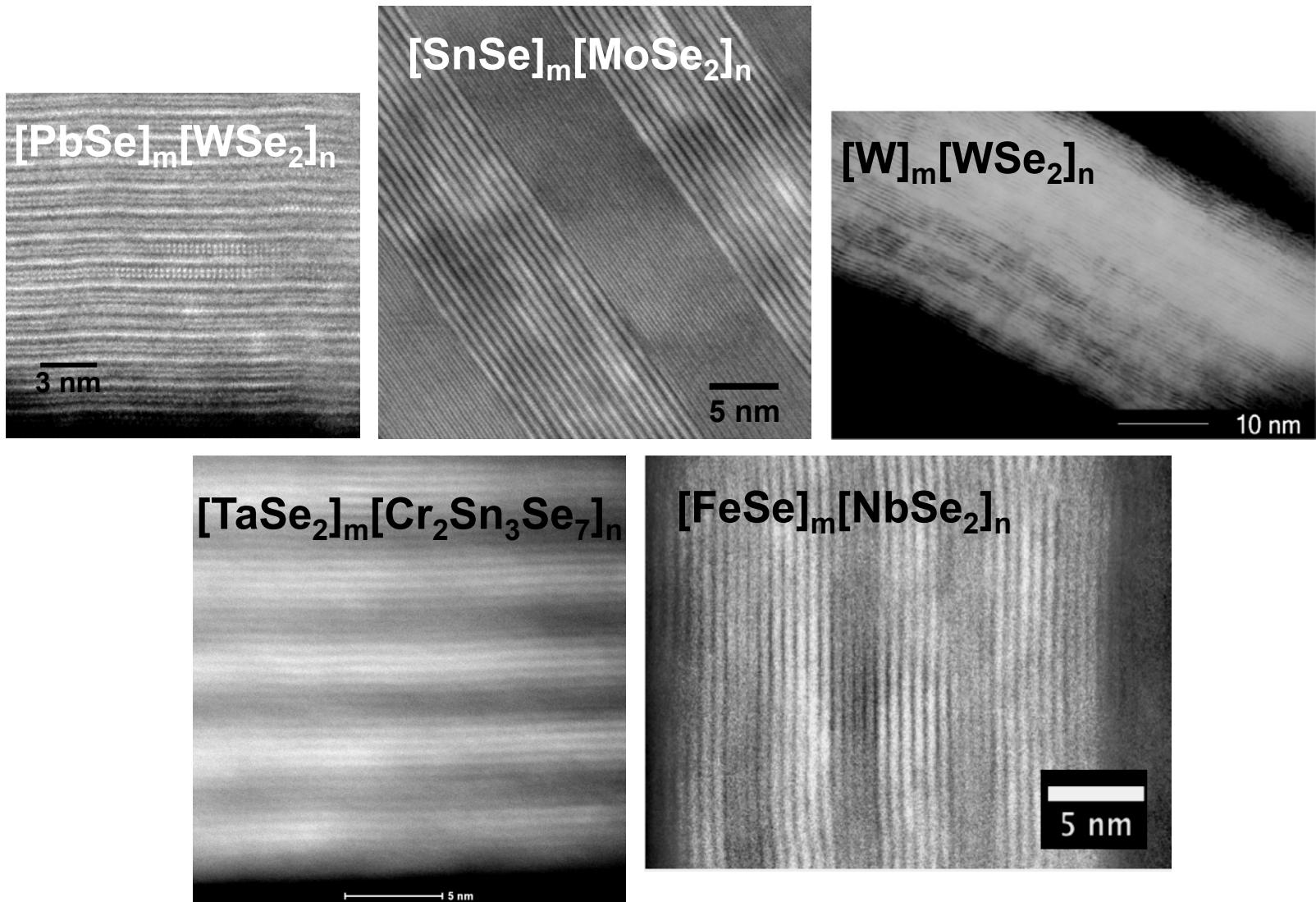
Over 200 isomers for  $n$  and  $m$   
6 or less

Over 20,000 isomers for  $n$   
and  $m$  10 or less



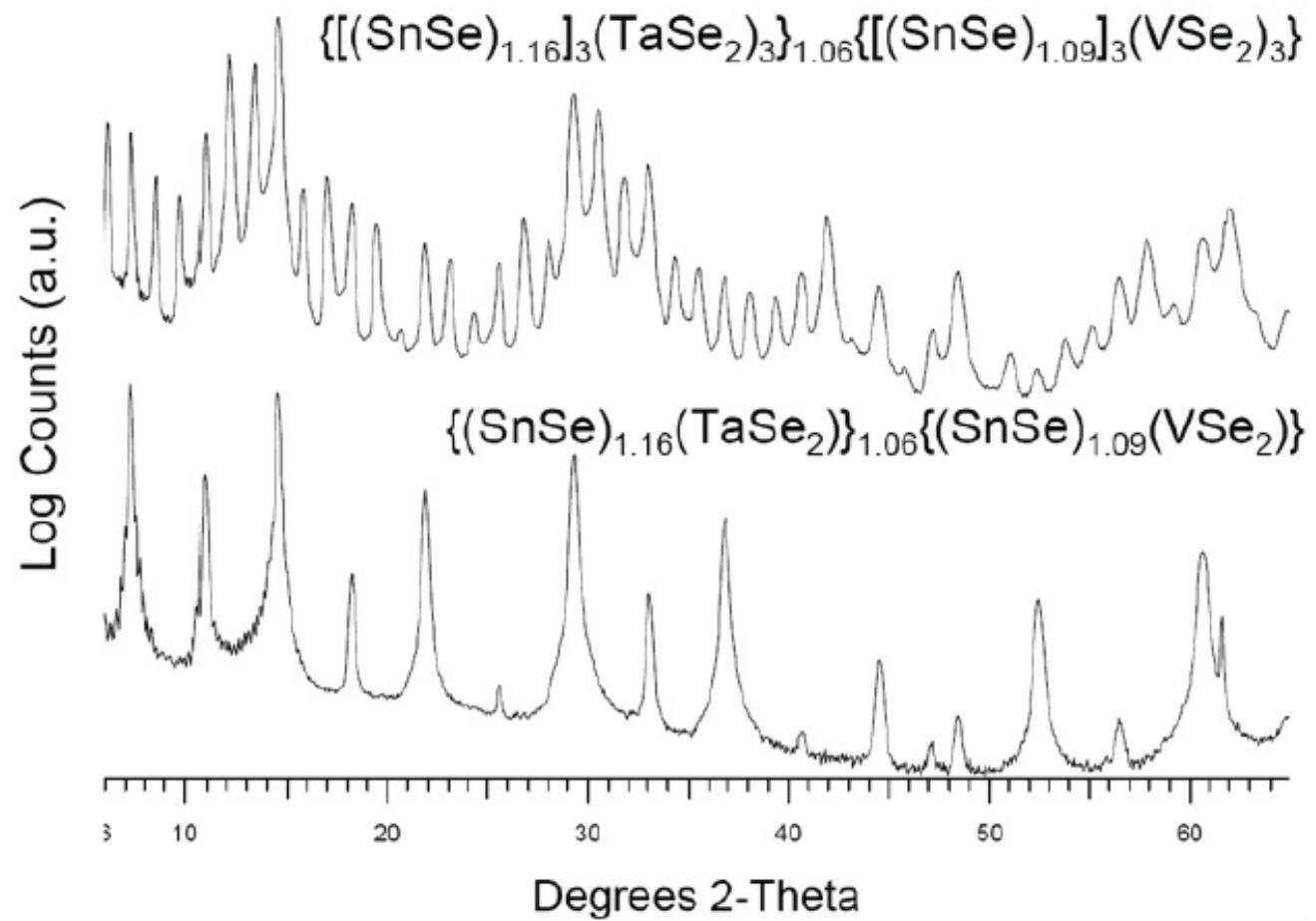
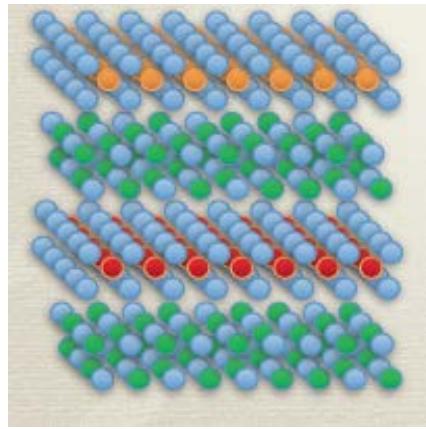
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# A general design principle – materials by design!



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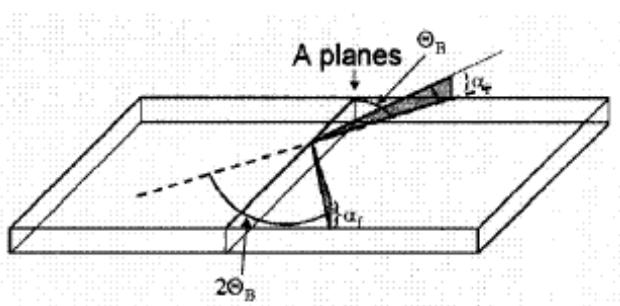
# Three component ferecrystals – precisely controlled modulation doping



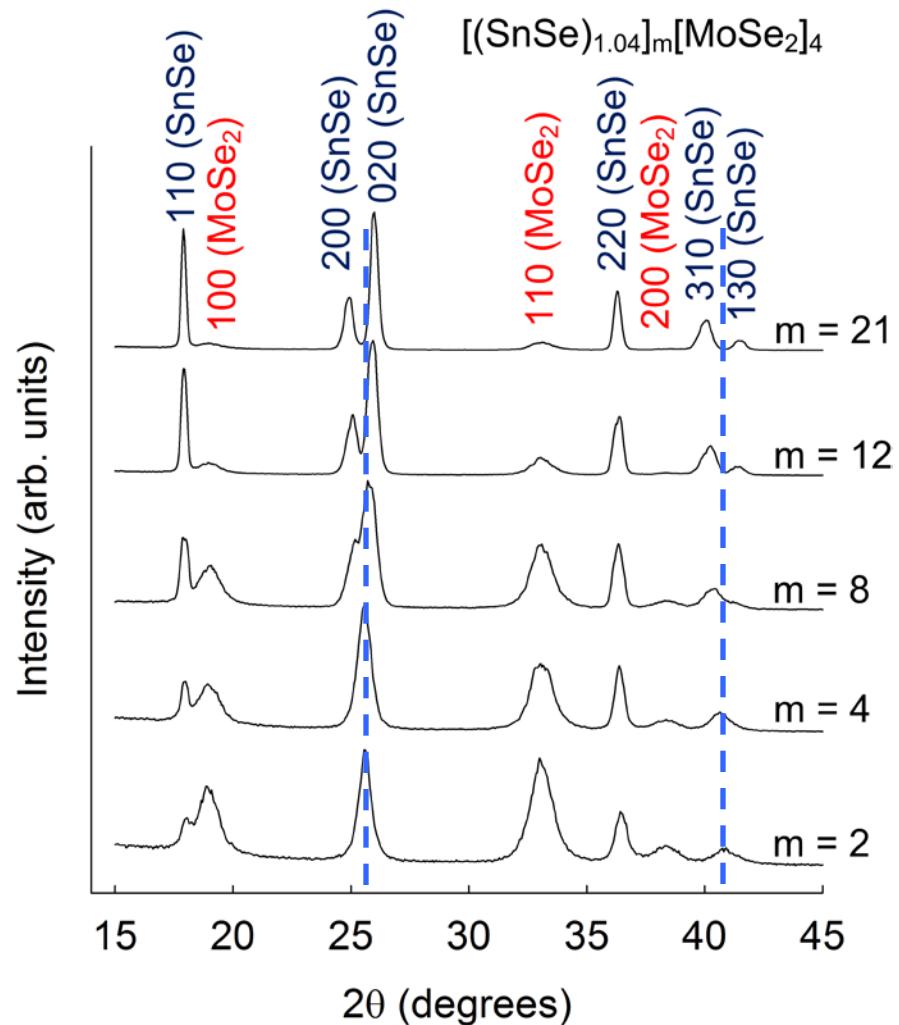
# Size-induced structural transformations

In-plane Synchrotron X-ray Diffraction

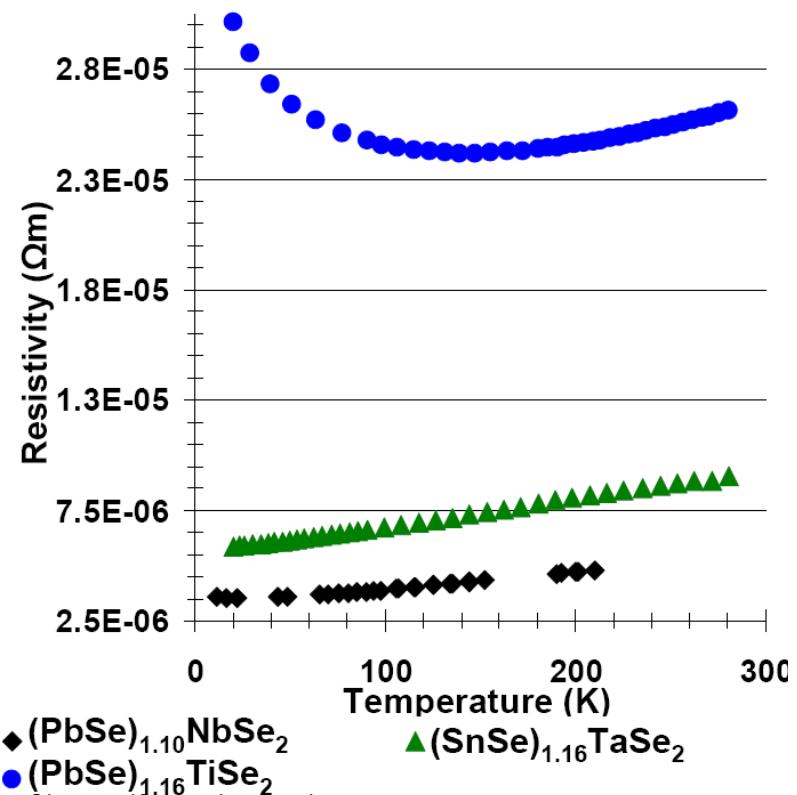
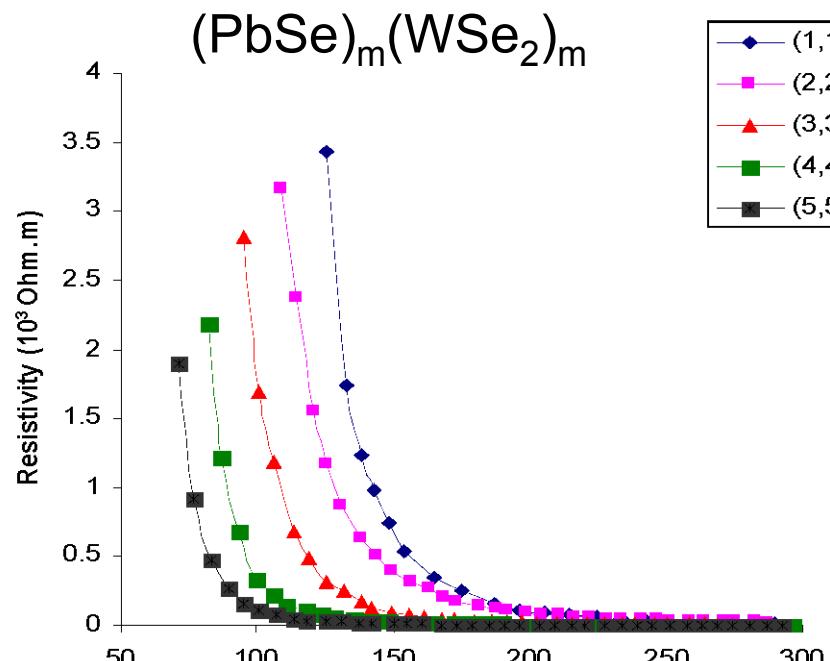
( $hk0$ ) reflections from individual components



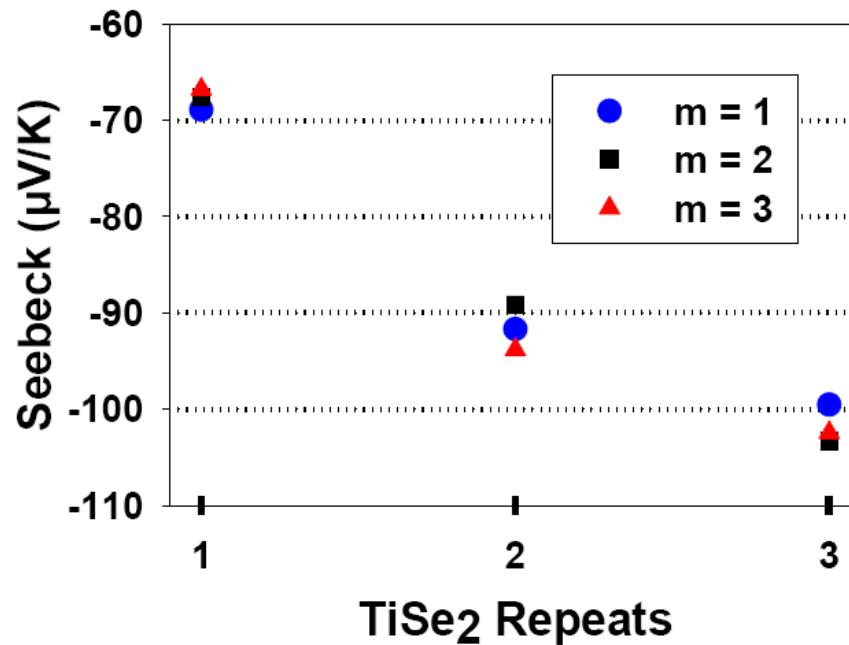
Beam line 33-BM-C, APS  
Dr. Paul Zschack



# Tunable electrical properties



# Seebeck systematically varies with number of $\text{TiSe}_2$ layers



**Misfit  $(\text{PbSe})_{1.16}(\text{TiSe}_2)_2$**   
**(N. Giang et al.)**

$(\text{PbSe})_{1.16}(\text{TiSe}_2)_2$  ferecrystal

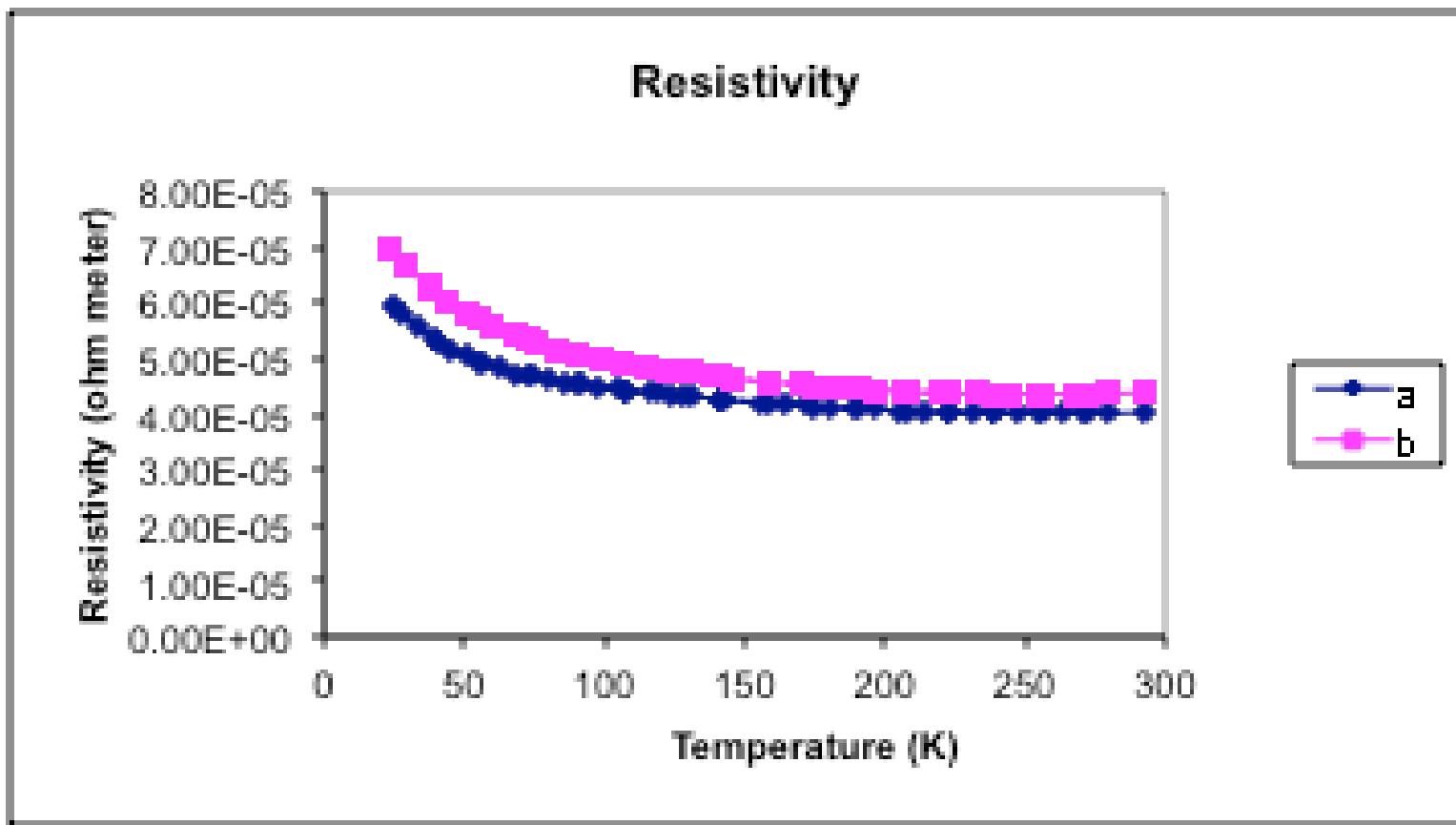
**Seebeck**  
 **$-50\mu\text{V}/\text{K}$**

**$-92\mu\text{V}/\text{K}$**

**Resistivity**  
 **$2 \times 10^{-4} \Omega\text{m}$**

**$3.7 \times 10^{-5} \Omega\text{m}$**

# Turbostratic TiSe<sub>2</sub>



Seebeck coefficients –  $a = -170 \mu\text{V/K}$   
 $b = -180 \mu\text{V/K}$

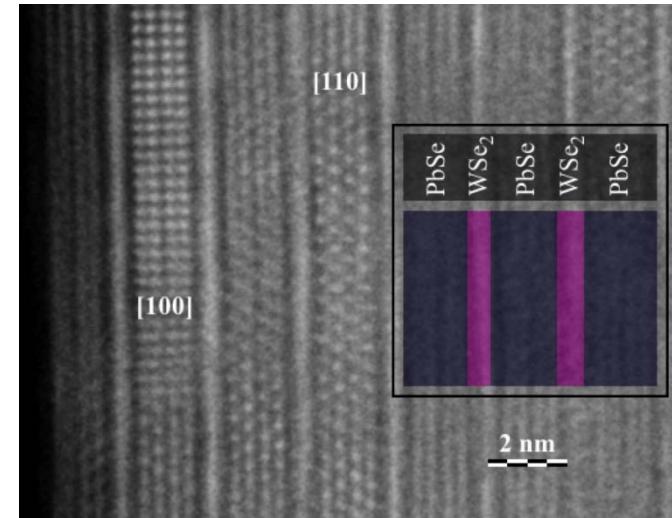
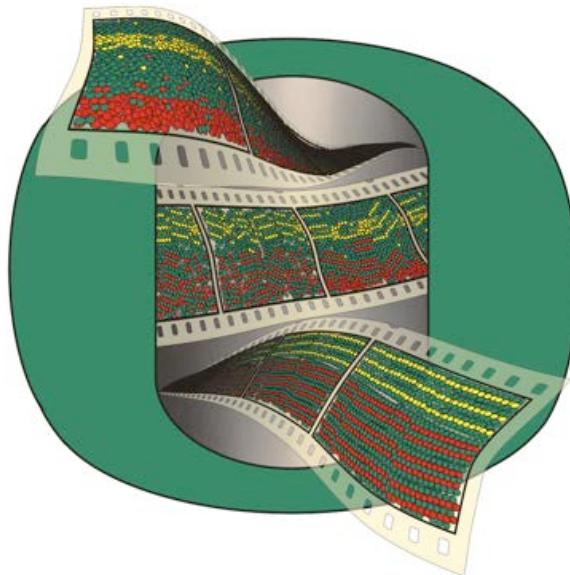
Unusual temperature dependence!



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# Summary

- Many new nanostructured ferecrysats can be synthesized.
- Structure of the constituent layers depends on layer thickness.
- Turbostratic disorder observed for all compounds investigated results in ultra-low thermal conductivity.
- Electrical properties depend on composition and nanostructure
- Power factor can be increased using nanostructure
  - Need to measure carrier concentrations



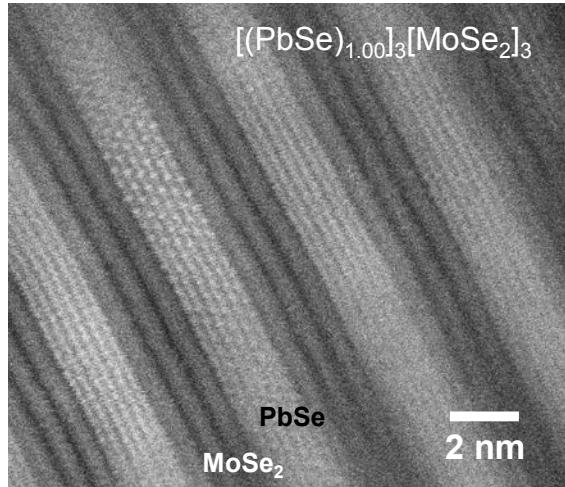
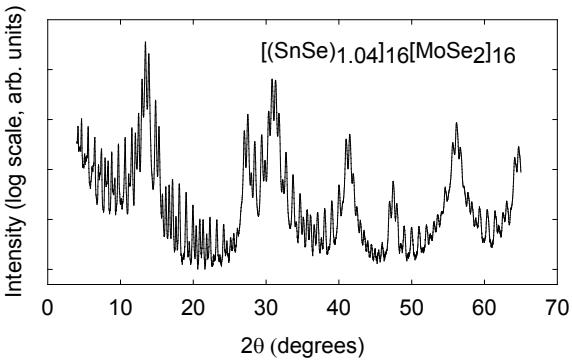
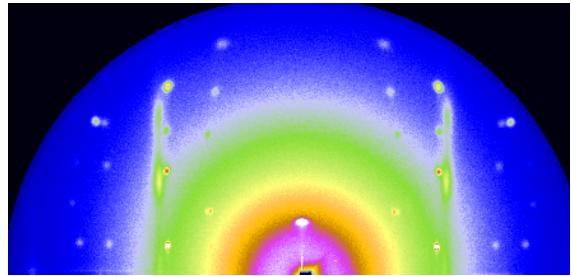
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Given a family of compounds  $(A)_m(B)_n(C)_p$ ,

Can one calculate thermal, electrical, and/or magnetic properties?

Can one calculate the trends in properties as a function of m, n, and p?

Can one suggest which structures A, B or C should be interleaved to achieve a particular property?



# Acknowledgements

Post doctoral researchers

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Colby Heideman  
Qiyin Lin



Collaborators:

Ian M. Anderson  
Wolfgang Neumann  
Paul Zschack



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